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Mechanical properties of Si₃N₄ matrix composites reinforced with SiC whiskers with oxide coatings

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Abstract—SiC whiskers were coated with MgO or TiO_2 by a sol-gel method in order to vary the interfacial strength between whisker and matrix. These whiskers with oxide coatings were used to fabricate Si_3N_4 matrix composites for the purpose of investigating the effects of the coating on mechanical properties such as fracture strength, fracture toughness, and fracture energy. The composite reinforced with whiskers with MgO coating showed higher fracture energy than that with TiO_2 coating, while the sample reinforced with whiskers with TiO_2 coating exhibited better fracture toughness and strength. These results suggest that MgO coating reduced the interfacial strength, while TiO_2 coating increased it.

Keywords: composite; coating; Si₃N₄; SiC; whisker.

1. INTRODUCTION

Ceramics are promising materials for high-temperature structural applications. However, their use is limited to only a few fields because of their low fracture toughness. Thus, much work has focused on ceramic matrix composites reinforced with fibers or whiskers to increase fracture toughness [1–4]. Whisker composite is a promising material, and it has been found to be effective for increasing the fracture toughness of Al₂O₃. Furthermore, Si₃N₄ matrix composites reinforced with SiC whiskers have also been investigated extensively [5–8], and the effects of whiskers on high temperature properties [9, 10] and fracture energy [11, 12] have been recognized.

Whisker alignment [13–17], and interfacial properties between whisker and matrix [13, 18–20], have been considered the most important factors in improving the fracture toughness of composites. In our previous work [21, 22], Ti and Al coatings on the whisker surface were found to be effective in increasing the fracture toughness of SiC whisker/Si₃N₄ composites because silicide was formed at the interface between whisker and matrix, whereas Mg or Y coatings, which formed nitride at the interface, were effective for increasing fracture energy. Metal coating was, however, a very time-consuming process, since it was done by RF-sputtering.

Therefore, in this study, a sol-gel method was used to coat MgO and TiO₂ on SiC whiskers because the coatings were expected to be formed in a shorter time. In order to clarify the effects of the coating, MgO and TiO₂, which were not used as sintering additives in this experiment, were selected from among the Mg, Y, Al, and Ti oxides. Then, Si₃N₄ matrix composites reinforced with SiC whisker coated with oxide were fabricated to investigate the effects of the coating on mechanical properties.

2. EXPERIMENTAL PROCEDURES

2.1. Sample preparation

2.1.1. Surface coating. Commercially available SiC whiskers (TWS-400 Tokai Carbon Co., Ltd, Japan) were used in the following investigations. In the case of MgO coating, Mg(CH₃OCH₂CH₂O)₂ (Soekawa Chemical Co., Ltd, Japan) as a metal alkoxide and CH₃OCH₂CH₂OH (Wako Chemical Co., Inc., Japan) as a solvent were used. In the case of the TiO₂ coating, Ti(OC₂H₅)₄ (Soekawa Chemical Co., Ltd, Japan) as a metal alkoxide, C₂H₅OH (Kanto Chemical Co., Inc., Tokyo, Japan) as a solvent, and HNO₃ (Koso Chemical Co., Ltd, Tokyo, Japan) as a catalyst were Each solvent was selected according to the alkoxide substituent. whiskers were put into the solvent and mixed by means of a magnetic stirrer to obtain a well-dispersed whisker suspension. The metal alkoxide was dissolved in the solvent in another vessel. Then, the SiC whisker suspension was added to the metal alkoxide solution and thoroughly mixed. After that, H₂O and the catalyst were added to the metal alkoxide solution with SiC. After the hydrolyzation of metal alkoxide, the whiskers with coatings were filtered and washed with the solvent. Up to the hydrolyzation step, all operations were carried out in dry N₂ to avoid the adsorption of water in the air. Subsequently, the filtered cake was redispersed, dried at 120°C for 3 h in air, and finally heat treated at 500°C for 5 h in air.

In the case of the MgO coating, the molar ratio of H_2O to $Mg(CH_3OCH_2CH_2O)_2$ was varied in the range of 0.1–0.4, and reaction temperature, reaction time, and alkoxide content were kept constant. In the case of the TiO_2 coating, four important parameters, namely, molar ratio of H_2O to $Ti(OC_2H_5)_4$, reaction temperature, reaction time, and alkoxide content, were varied because a uniform TiO_2 coating was more difficult to form than with MgO.

2.1.2. Whisker-reinforced composite. Commercial-grade Si_3N_4 powder (SN-E10, Ube Industries Ltd, Japan) with 5 wt% Y_2O_3 (99.9%, Shin-Etsu Chemical Co., Ltd, Japan) and 2 wt% Al_2O_3 (99.9%, Plax Air Surfaces Technology Inc., USA), as sintering additives, and 20 wt% SiC whiskers with oxide coatings were mixed by ball-milling for 16 h using *n*-butanol as the liquid medium. These powder mixtures were hot-pressed in a carbon die at a temperature of 1800° C under 30 MPa for 1 h in a 0.1 MPa N_2 gas.

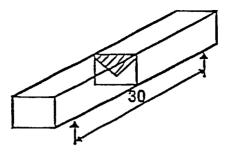


Figure 1. Schematic drawing of CN test piece.

2.2. Evaluation

Rectangular bars, $3 \times 4 \times 40$ mm, were machined from the hot-pressed billets for measuring mechanical properties. Fracture strength of the whisker composites was evaluated by a 3-point bending test with a 30-mm span in accordance with JIS-R1601. Fracture toughness was measured by a 4-point bending test, with a 30-mm outer span and a 10-mm inner span, using single edge-notched beam (SENB) specimens. The notch width and depth were 0.1 and 0.75 mm, respectively. The fracture toughness measured by the SENB method is overestimated when the notch width is greater than the critical value. However, it has been reported that the fracture toughness of Si₃N₄ matrix composites reinforced with SiC whiskers obtained using a 0.1-mm notch width is set due value on [11]. These tests were performed at a displacement rate of 0.5 mm/min. Fracture energy was obtained from the accurate measurement of the load point displacements by a linear variable differential transducer (LVDT) [11] for chevron-notched (CN) specimens in a 3-point bending test. The size of the CN specimen is shown in Fig. 1. This test was conducted at a displacement rate of 0.05 mm/min. Under these conditions, Si₃N₄ matrix composites reinforced with SiC whiskers are fractured stably [11, 12]. Five bars were used to measure each mechanical property.

The whisker surfaces were examined with a scanning electron microscope (SEM; T300A, JEOL Ltd, Tokyo, Japan) and a transmission electron microscope (TEM; CX200, JEOL Ltd, Tokyo, Japan). Compositions of the coating layer were analyzed by energy dispersive spectroscope (EDS; Noran Instrument, Tokyo, Japan). The thickness of the coating layer was measured from the TEM observations. The fractured surfaces of the CN specimens were investigated by SEM.

3. RESULTS AND DISCUSSION

3.1. Whisker coating

An MgO coating layer was easily obtained at an H₂O/Mg(CH₃OCH₂CH₂O)₂ molar ratio of 0.4. Figure 2 shows a transmission electron micrograph of the whisker surface for the case of an H₂O/Mg(CH₃OCH₂CH₂O)₂ ratio of 0.4. The EDS spectrum

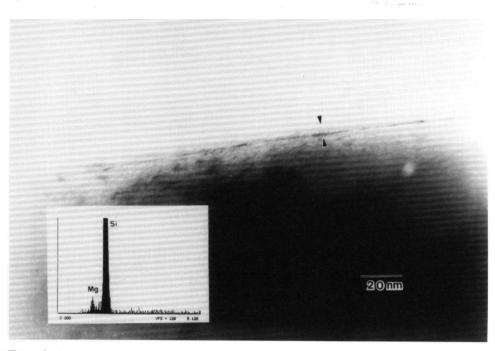


Figure 2. TEM micrograph of whisker surface with MgO coating and EDS spectrum acquired from the coating.

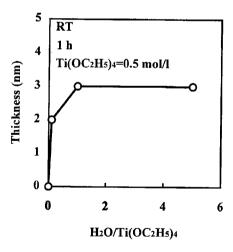


Figure 3. Relation between coating thickness and molar ratio of H₂O/Ti(OC₂H₅)₄.

obtained from the coating is inset in Fig. 2. It was confirmed that the coating layer included Mg element and that the thickness was 10-12 nm.

Figure 3 shows the relationship between the thickness and the concentration ratio of $H_2O/Ti(OC_2H_5)_4$. The coating experiment was conducted with a $Ti(OC_2H_5)_4$ content

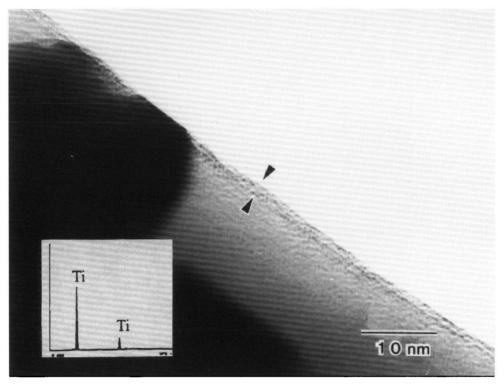
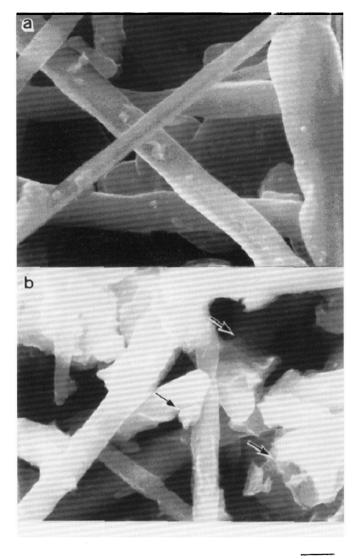


Figure 4. TEM micrograph of whisker surface with TiO₂ coating and EDS spectrum acquired from the coating.

of 0.5 mol/l at room temperature for 1 h. 3-μm-thick coatings were formed at the H₂O/Ti(OC₂H₅)₄ ratios above 0.1, while no coating was obtained at an H₂O/Ti (OC₂H₅)₄ ratio of 0.01. Figure 4 shows a transmission electron micrograph of a whisker surface for the case of an H₂O/Ti(OC₂H₅)₄ ratio of 0.1. The EDS spectrum obtained from the coating is inset in Fig. 4. It was confirmed that the coating layer included Ti element and was 2-3 nm thick. Figure 5 shows scanning electron micrographs of whiskers after drying at H₂O/Ti(OC₂H₅)₄ ratios of 0.1 (Fig. 5a) and 5.0 (Fig. 5b). When the $H_2O/T_1(OC_2H_5)_4$ ratio was 5.0, many large particles were observed between whiskers, as shown by the arrows in Fig. 5b. If the molar ratio of alkoxide to water was less than 4, linear polymers were apt to form. When the ratio was above 4, three-dimensional polymers or spherical colloids were prone to form. Linear polymers dehydrated with -OH of the whisker surface to form a coating layer. On the other hand, three-dimensional polymers or spherical colloids tended to peel off even if they dehydrated with -OH of the whisker surface since they were much larger in size than linear polymers. As a result, they seemed to polymerize with each other to achieve larger sizes and remained between whiskers because of the high concentration of water even after washing [23-25].



1 µm

Figure 5. SEM micrographs of whisker surface with TiO_2 coating: (a) molar ratio of $H_2O/Ti(OC_2H_5)_4 = 0.1$, (b) molar ratio of $H_2O/Ti(OC_2H_5)_4 = 5.0$.

Figures 6 and 7 show the relationships between thickness and reaction temperature and between thickness and reaction time, respectively. These experiments were performed at an $H_2O/Ti(OC_2H_5)_4$ ratio of 1.0 and a $Ti(OC_2H_5)_4$ content of 0.5 mol/l. It was found that the reaction temperature and time have no effect on the coating thickness under these conditions. We hypothesize that $Ti(OC_2H_5)_4$ was so reactive that it could hydrolyze sufficiently at room temperature in a shorter time.

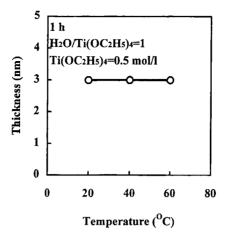


Figure 6. Relationship between coating thickness and reaction temperature.

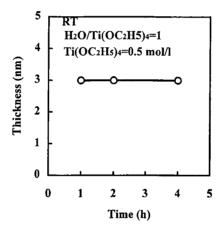


Figure 7. Relationship between coating thickness and reaction time.

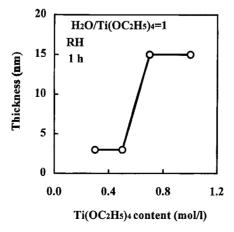


Figure 8. Relationship between coating thickness and Ti(OC₂H₅)₄ content.

Figure 8 shows the relationship between the thickness and $Ti(OC_2H_5)_4$ content. The thickness increased steeply to 15 nm in the $Ti(OC_2H_5)_4$ content range of 0.5 to 0.7 mol/l, while it remained constant at about 3 nm in the range of 0.3 to 0.5 mol/l. When the $Ti(OC_2H_5)_4$ content was 1.0 mol/l, many large particles existed around the whiskers, which is similar to the case shown in Fig. 5b. We hypothesize that Ti alkoxides reacted with each other to achieve large sizes and peeled off even if they dehydrated with -OH of the whisker surface.

3.2. Mechanical properties

The mechanical properties of the composites are shown in Table 1. Although there were differences in the data, they were significant because the standard deviation was very small. It was found that in the case of the MgO coating, the bending strength and fracture toughness were somewhat low, and fracture energy was high, while in the case of the TiO₂ coating, the bending strength and fracture toughness were relatively high, and fracture energy was low.

Figure 9 shows the SEM micrographs of the fractured surfaces of CN test pieces reinforced with: (a) as-received whiskers, (b) MgO-coated whiskers, and (c) TiO₂coated whiskers. Careful examination of Fig. 9 revealed whisker pull-out, as indicated by the arrows, and the number of whiskers pulled-out was counted. The results are also shown in Table 1. The number of pulled-out whiskers changed according to the coating oxide. In the sample with MgO-coated whiskers, more whiskers were pulled-out than in the samples with TiO₂-coated whiskers. The number of the pulled-out whiskers with TiO₂ coating was less than that with no coating. The increase in fracture energy seemed to be related to the whisker pull-out since it showed the same trend as the number of the pulled-out whiskers. Because whiskers are pulled out after a crack passes by, the increase in fracture energy is thought to correspond to the increase of fracture resistance to the crack propagation. The results concerning the effects of whisker coatings on the mechanical properties and the trend in the number of pulled-out whiskers are similar to the results obtained for metal coatings [21, 22]. Therefore, we propose that MgO coating, which seemed to form nitride at the interface between whisker and matrix, reduces the interfacial strength and TiO₂ coating, which seemed to form silicide at the interface, enhances it.

Table 1. Mechanical properties of composites

Coating	Bending strength (MPa)	Fracture toughness (MPa m ^{1/2})	Fracture energy $(J m^{-2})$	Number of whiskers (mm ⁻²)
none	1080 ± 20	8.7 ± 0.1	161 ± 5	3600
MgO	990 ± 40	8.2 ± 0.1	175 ± 6	6000
TiO ₂	1130 ± 20	9.0 ± 0.1	150 ± 3	3000



Figure 9. SEM micrographs of fracture surfaces for CN specimens reinforced with: (a) as-received whiskers, (b) MgO-coated whisker, and (c) TiO₂-coated whisker.

4. CONCLUSIONS

The following conclusions were reached based on the results of this investigation.

- 1) In the case of the MgO coating, bending strength and fracture toughness were somewhat low, and fracture energy was high. On the other hand, in the case of the TiO₂ coating, bending strength and fracture toughness were relatively high, and fracture energy was low.
- 2) MgO coating on the whisker surface has an effect on the mechanical properties of composites that is similar to that of Mg, and TiO₂ coating has an effect that is similar to that of Ti.
- 3) TiO₂ coating on the whisker increased the interfacial strength between whisker and matrix, while MgO coating reduced it.

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